



SI EDITION

# PHYSICAL METALLURGY PRINCIPLES

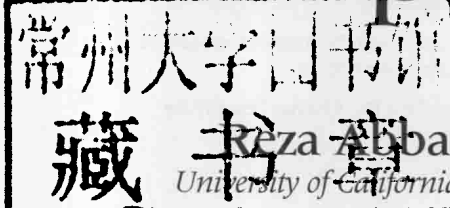
Fourth Edition

REZA ABBASCHIAN  
LARA ABBASCHIAN  
ROBERT E. REED-HILL



FOURTH EDITION, SI

# Physical Metallurgy Principles



Reza Abbaschian  
*University of California, Riverside*

Lara Abbaschian  
*The Boston Consulting Group*

Robert E. Reed-Hill



CENGAGE  
Learning™

Australia • Brazil • Japan • Korea • Mexico • Singapore • Spain • United Kingdom • United States

**Physical Metallurgy Principles,  
Fourth Edition, SI**

**Reza Abbaschian, Lara Abbaschian,  
Robert E. Reed-Hill**

Director, Global Engineering Program:  
Chris Carson

Senior Developmental Editor: Hilda Gowans

Development Editor: Swati Meherishi

Editorial Assistant: Jennifer Dinsmore

Associate Marketing Manager: Lauren Betsos

Senior Content Project Manager:  
Kim Kusnerak

Production Service: RPK Editorial Services

Compositor: Integra

Senior Art Director: Michelle Kunkler

Internal Designer: JenzDesign

Cover Designer: Andrew Adams

Cover Image: © Shannon West/Shutterstock  
Images LLC

Text and Image Permissions Researcher:  
Natalie Barrington

Senior First Print Buyer: Doug Wilke

© 2010, 1994 Cengage Learning

ALL RIGHTS RESERVED. No part of this work covered by the copyright herein may be reproduced, transmitted, stored, or used in any form or by any means—graphic, electronic, or mechanical, including but not limited to photocopying, recording, scanning, digitizing, taping, Web distribution, information networks, information storage and retrieval systems, or in any other manner—except as may be permitted by the license terms herein.

For product information and technology assistance, contact us at  
**Cengage Learning Customer & Sales Support, 1-800-354-9706.**

For permission to use material from this text or product,  
submit all requests online at [www.cengage.com/permissions](http://www.cengage.com/permissions).  
Further permissions questions can be emailed to  
[permissionrequest@cengage.com](mailto:permissionrequest@cengage.com).

Library of Congress Control Number: 2009924307

ISBN-13: 978-0-495-43851-9

ISBN-10: 0-495-43851-0

**Cengage Learning**

200 First Stamford Place, Suite 400

Stamford, CT 06902

USA

Cengage Learning is a leading provider of customized learning solutions with office locations around the globe, including Singapore, the United Kingdom, Australia, Mexico, Brazil, and Japan. Locate your local office at: [international.cengage.com/region](http://international.cengage.com/region).

Cengage Learning products are represented in Canada by  
Nelson Education Ltd.

For your course and learning solutions, visit  
[www.cengage.com/engineering](http://www.cengage.com/engineering).

Purchase any of our products at your local college store or at our preferred  
online store [www.ichapters.com](http://www.ichapters.com).



# Preface to the SI Edition

This edition of *Physical Metallurgy Principles* has been adapted to incorporate the International System of Units (*Le Système International d'Unités* or SI) throughout the book.

**LE SYSTÈME INTERNATIONAL D'UNITÉS** The United States Customary System (USCS) of units uses FPS (foot–pound–second) units (also called English or Imperial units). SI units are primarily the units of the MKS (meter–kilogram–second) system. However, CGS (centimeter–gram–second) units are often accepted as SI units, especially in textbooks.

**USING SI UNITS IN THIS BOOK** In this book, we have used both MKS and CGS units. USCS units or FPS units used in the US Edition of the book have been converted to SI units throughout the text and problems. However, in case of data sourced from handbooks, government standards, and product manuals, it is not only extremely difficult to convert all values to SI, it also encroaches upon the intellectual property of the source. Also, some quantities such as the ASTM grain size number and Jominy distances are generally computed in FPS units and would lose their relevance if converted to SI. Some data in figures, tables, examples, and references, therefore, remains in FPS units. For readers unfamiliar with the relationship between the FPS and the SI systems, conversion tables have been provided inside the front and back covers of the book.

To solve problems that require the use of sourced data, the sourced values can be converted from FPS units to SI units just before they are to be used in a calculation. To obtain standardized quantities and manufacturers' data in SI units, the readers may contact the appropriate government agencies or authorities in their countries/regions.

**INSTRUCTOR RESOURCES** A Printed Instructor's Solutions Manual in SI units is available on request. An electronic version of the Instructor's Solutions Manual, and PowerPoint slides of the figures from the SI text are available through [www.cengage.com/engineering](http://www.cengage.com/engineering).

The readers' feedback on this SI Edition will be most appreciated and will help us improve subsequent editions.

**The Publishers**



# Preface

**THE FIRST EDITION** In recent years, introductory physical metallurgy textbooks have attempted to achieve three goals: to explain basic metallurgical phenomena, to identify the compositions and properties of commercial alloys, and to teach principles of metal fabrication. Because all three phases are generally covered in a single course of one or two semesters, none of them receives adequate treatment. A natural question that arises is which of the three is most generally important to the engineering student. In this regard, it should be pointed out that metal fabrication and alloy properties are fields that are characteristically factual in nature. Although, strong arguments can be presented for including these areas of study, it must still be admitted that time spent learning large numbers of apparently unrelated facts is frequently wasted. Information of this sort is easily forgotten and, what is more, today's alloys and methods are not necessarily those of tomorrow. On the other hand, the theoretical approach to physical metallurgy is premised on the belief that the properties of metals and alloys are determined by simple physical laws, and that it is not necessary to consider each alloy as a separate entity. Recent advances in the physics and chemistry of metals have gone far toward finding the needed interrelations.

This book is intended for use as an introductory course (of one or two semesters) in physical metallurgy and is designed for all engineering students at the junior or senior level. A number of chapters dealing with advanced topics, such as Chapters 10, 11, 15, and 19 may be omitted in their entirety when the book is used for a one-semester course. Prerequisites are college physics, chemistry, and strength of materials. An engineering course in thermodynamics or physical chemistry is also considered desirable but not essential. The approach is largely theoretical, but all major phases of metal behavior normally found in physical metallurgy textbooks are covered. In this respect, statistical mechanics and dislocation theory are used to explain plastic deformation and thermal effects in metals. Vacancies are treated in some detail because their study may be used to obtain a true appreciation for the meaning of activation energies in metals. Deformation twinning is given considerable attention not only because this type of deformation has become increasingly more important, but also because twinning theory leads directly into the important subject area of martensite transformations.

On the whole, it is believed that the treatment used in this book is in harmony with current trends toward a more fundamental approach in engineering education.

The author would like to acknowledge that the lectures of Dr. A. S. Nowick and Dr. W. D. Robertson at the Hammond Laboratory, Yale University, were largely instrumental in inspiring the writing of this book. The helpful suggestions from Dr. F. N. Rhines on the subject of creep are also gratefully acknowledged.

**THE SECOND EDITION** The basic plan and philosophy of the original edition are continued in this volume. The major changes in the new edition are largely the

result of constructive suggestions and advice by Professor Richard W. Heckel, of Drexel University, Dean Walter S. Owen, of Northwestern University, and Professor Marvin Metzger, of the University of Illinois. One result of these suggestions is the inclusion of a chapter on nucleation and growth kinetics. The outline of this chapter was also inspired by a set of class notes kindly loaned to the author by Professor Heckel. The considerable assistance of Dr. John Kronsbein in revising and expanding Chapter 3, Elementary Theory of Metals, is also gratefully acknowledged.

As a consequence of requests for the inclusion of topics either missing or too lightly covered in the first edition, the new book has been increased in size by about ten percent. In a broad sense, the additional material fits into two classifications. First are the topics that have recently become significant in the field of metallurgy. The second group consists of well-established subjects not covered in the first edition, but which, from use of the text, were found to be needed for a more unified presentation. Among the former subject areas are electron microscopy, fracture mechanics, superconductivity, superplasticity, dynamic recovery, dynamic strain aging, electrotransport, thermal migration, and emissary dislocations. In the latter category belong the new chapter on nucleation and growth kinetics and such topics as magnetism, the zone theory of alloy phases, the five degrees of freedom of a grain boundary, the phase rule, true stress and true strain, coring and homogenization of castings, work hardening, and diffusion in nonisomorphic systems.

The number of problems is substantially increased over that in the original book, in conformity with the current trend in engineering to place more emphasis on problem solving. Problems have been written with the aim of both illustrating points covered in the text and exposing the student to material and concepts not covered directly in the book.

The helpful assistance of Dr. John Hren, Dr. Robert T. DeHoff, Dr. Derek Dove, Dr. Ellis Verink, and Dr. E. N. Rhines, all of the University of Florida, who either reviewed sections of the book or gave suggestions, is acknowledged with thanks.

**THE THIRD EDITION** The basic philosophy underlying the original edition is retained in the third edition. However, a number of significant improvements have been incorporated in the third edition. The International System of Units is now employed throughout text and problems. A chapter devoted to important non-ferrous metal has been added. Fracture mechanics is covered in much greater depth and breadth in a separate chapter. The treatment of solidification has been expanded and brought up to date and includes an extensive coverage of liquid metals as well as the Scheil equation and eutectic freezing. The section on the transmission electron microscope has been expanded and a detailed discussion of the scanning electron microscope has been added. Grain boundaries are now covered in a separate chapter that includes coincident site boundaries. The subject of dislocations has been reorganized and consolidated. Chapter 4 considers the geometrical aspects of dislocations while Chapter 5 treats the relationship of dislocations to plastic deformation. The phase diagrams in the text have been brought up to date. In the steel chapters, the transformations of austenite to pearlite, bainite, and martensite, and the tempering of martensite have been modernized. In the deformation twinning and martensite reactions chapter, less emphasis is placed on twinning phenomena per se while the role that twinning can play in the plastic deformation of polycrystalline metals has been added. In the martensite section, thermoelastic deformation and shape memory effects are now covered.



The authors would like to thank Professor William C. Leslie of the University of Michigan and Professor Daniel N. Beshers of Columbia University for their extensive and constructive suggestions concerning material that needed to be corrected or added to the third edition.

The authors would also like to acknowledge the assistance of Professors Paul C. Holloway and Rolf N. Hummel, University of Florida. We are also greatly indebted to all of the following for their constructive reviews of the manuscript of the third edition: Professor William A. Jesser, University of Virginia; Professor William G. Ovens, Rose-Hulman Institute of Technology; Professor Dale E. Wittmer, Southern Illinois University; Professor James C. M. Li, University of Rochester; Professor Alan R. Pelton, University of Notre Dame; and Professor Samuel J. Hruska, Purdue University.

**THE FOURTH EDITION** The original philosophy of the former editions has been kept in this fourth edition. The text retains its easy-to-read format so that the essence of the information is most successfully communicated. Recent references have been incorporated to complement the remaining original references which provide historic context. Indeed, with electronic search engines limited to more recently published literature, identification and acknowledgment of the pioneers of the field often gets overlooked. To increase the focus of the text, Chapters 22 and 23 were removed, with salient portions incorporated into other chapters. As such, the book is more adaptable for a one or two semester introductory course for juniors or seniors in materials science and engineering. New figures were added to enhance understanding of the text. Sections on nano-structures have been included to demonstrate the applicability of the physical metallurgy principles to current nanotechnology trends; however, the authors leave it to the readers to independently pursue the area more thoroughly.

The authors would like to thank the reviewers whose constructive reviews of the third edition drove the aforementioned changes: Professor Richard B. Griffin, Texas A & M University; Professor Dong-Joo (Daniel) Kim, Auburn University; Professor Anthony P. Reynolds, University of South Carolina; Professor Christopher A. Schuh, Massachusetts Institute of Technology; Professor Jiahong Zhu, Tennessee Technical University. Special thanks to Hilda Gowans, Senior Development Editor, whose assistance was invaluable throughout the revisions, to Chris Carson and the Engineering group at Cengage Learning and to Rose Kernan who managed the production of this new edition. We also thank Dr. Abraham Munitz for providing the pictures for the chapter headings. Finally, we again acknowledge the critical inputs and contributions of all who are listed in the prefaces of the earlier editions.

Robert E. Reed-Hill  
Reza Abbaschian  
Lara Abbaschian



# Contents

---

## CHAPTER 1 THE STRUCTURE OF METALS

1

**1.1** The Structure of Metals, 1 **1.2** Unit Cells, 2 **1.3** The Body-Centered Cubic Structure (BCC), 3 **1.4** Coordination Number of the Body-Centered Cubic Lattice, 4 **1.5** The Face-Centered Cubic Lattice (FCC), 4 **1.6** The Unit Cell of the Hexagonal Closed-Packed (HCP) Lattice, 5 **1.7** Comparison of the Face-Centered Cubic and Close-Packed Hexagonal Structures, 6 **1.8** Coordination Number of the Systems of Closest Packing, 7 **1.9** Anisotropy, 7 **1.10** Textures or Preferred Orientations, 8 **1.11** Miller Indices, 9 **1.12** Crystal Structures of the Metallic Elements, 14 **1.13** The Stereographic Projection, 15 **1.14** Directions that Lie in a Plane, 16 **1.15** Planes of a Zone, 17 **1.16** The Wulff Net, 17 **1.17** Standard Projections, 21 **1.18** The Standard Stereographic Triangle for Cubic Crystals, 24 Problems, 26 References, 28

---

## CHAPTER 2 CHARACTERIZATION TECHNIQUES

29

**2.1** The Bragg Law, 30 **2.2** Laue Techniques, 33 **2.3** The Rotating-Crystal Method, 35 **2.4** The Debye-Scherrer or Powder Method, 36 **2.5** The X-Ray Diffractometer, 39 **2.6** The Transmission Electron Microscope, 40 **2.7** Interactions Between the Electrons in an Electron Beam and a Metallic Specimen 46 **2.8** Elastic Scattering, 46 **2.9** Inelastic Scattering, 46 **2.10** Electron Spectrum, 48 **2.11** The Scanning Electron Microscope, 48 **2.12** Topographic Contrast, 50 **2.13** The Picture Element Size, 53 **2.14** The Depth of Focus, 54 **2.15** Microanalysis of Specimens, 55 **2.16** Electron Probe X-Ray Microanalysis, 55 **2.17** The Characteristic X-Rays, 56 **2.18** Auger\* Electron Spectroscopy (AES), 58 **2.19** The Scanning Transmission Electron Microscope (STEM), 60 Problems, 60 References 61

---

## CHAPTER 3 CRYSTAL BINDING

62

**3.1** The Internal Energy of a Crystal, 62 **3.2** Ionic Crystals, 62 **3.3** The Born Theory of Ionic Crystals, 63 **3.4** Van Der Waals Crystals, 68 **3.5** Dipoles, 68 **3.6** Inert Cases, 69 **3.7** Induced Dipoles, 70 **3.8** The Lattice Energy of an Inert-Gas Solid, 71 **3.9** The Debye Frequency, 72 **3.10** The Zero-Point Energy, 73 **3.11** Dipole-Quadrupole and Quadrupole-Quadrupole Terms, 75 **3.12** Molecular Crystals, 75 **3.13** Refinements to the Born Theory of Ionic Crystals, 75 **3.14** Covalent and Metallic Bonding, 76 Problems 80 References, 81



<b>CHAPTER 4</b>	<b>INTRODUCTION TO DISLOCATIONS</b>	<b>82</b>
	4.1 The Discrepancy Between the Theoretical and Observed Yield Stresses of Crystals, 82 4.2 Dislocations, 85 4.3 The Burgers Vector, 93 4.4 Vector Notation for Dislocations, 95 4.5 Dislocations in the Face-Centered Cubic Lattice, 96 4.6 Intrinsic and Extrinsic Stacking Faults in Face-Centered Cubic Metals, 101 4.7 Extended Dislocations in Hexagonal Metals, 102 4.8 Climb of Edge Dislocations, 102 4.9 Dislocation Intersections, 104 4.10 The Stress Field of a Screw Dislocation, 107 4.11 The Stress Field of an Edge Dislocation, 109 4.12 The Force on a Dislocation, 111 4.13 The Strain Energy of a Screw Dislocation, 114 4.14 The Strain Energy of an Edge Dislocation, 115 Problems, 116 References, 118	
<b>CHAPTER 5</b>	<b>DISLOCATIONS AND PLASTIC DEFORMATION</b>	<b>119</b>
	5.1 The Frank-Read Source, 119 5.2 Nucleation of Dislocations, 120 5.3 Bend Gliding, 123 5.4 Rotational Slip, 125 5.5 Slip Planes and Slip Directions, 127 5.6 Slip Systems, 129 5.7 Critical Resolved Shear Stress, 129 5.8 Slip on Equivalent Slip Systems, 133 5.9 The Dislocation Density, 133 5.10 Slip Systems in Different Crystal Forms, 133 5.11 Cross-Slip, 138 5.12 Slip Bands, 141 5.13 Double Cross-Slip, 141 5.14 Extended Dislocations and Cross-Slip, 143 5.15 Crystal Structure Rotation During Tensile and Compressive Deformation, 144 5.16 The Notation for the Slip Systems in the Deformation of FCC Crystals, 147 5.17 Work Hardening, 149 5.18 Considère's Criterion, 150 5.19 The Relation Between Dislocation Density and the Stress, 151 5.20 Taylor's Relation, 153 5.21 The Orowan Equation, 153 Problems, 154 References, 157	
<b>CHAPTER 6</b>	<b>ELEMENTS OF GRAIN BOUNDARIES</b>	<b>158</b>
	6.1 Grain Boundaries, 158 6.2 Dislocation Model of a Small-Angle Grain Boundary, 159 6.3 The Five Degrees of Freedom of a Grain Boundary, 161 6.4 The Stress Field of a Grain Boundary, 162 6.5 Grain-Boundary Energy, 165 6.6 Low-Energy Dislocation Structures, LEDS, 167 6.7 Dynamic Recovery, 170 6.8 Surface Tension of the Grain Boundary, 172 6.9 Boundaries Between Crystals of Different Phases, 175 6.10 The Grain Size, 178 6.11 The Effect of Grain Boundaries on Mechanical Properties: Hall-Petch Relation, 180 6.12 Grain Size Effects in Nanocrystalline Materials, 182 6.13 Coincidence Site Boundaries, 185 6.14 The Density of Coincidence Sites, 186 6.15 The Ranganathan Relations, 186 6.16 Examples Involving Twist Boundaries, 187 6.17 Tilt Boundaries, 189 Problems, 192 References, 193	
<b>CHAPTER 7</b>	<b>VACANCIES</b>	<b>194</b>
	7.1 Thermal Behavior of Metals, 194 7.2 Internal Energy, 195 7.3 Entropy, 196 7.4 Spontaneous Reactions, 196 7.5 Gibbs Free Energy, 197	

7.6 Statistical Mechanical Definition of Entropy, 199 7.7 Vacancies, 203  
 7.8 Vacancy Motion, 209 7.9 Interstitial Atoms and Divacancies, 211 Problems,  
 214 References, 215

<b>CHAPTER 8</b>	<b>ANNEALING</b>	<b>216</b>
	8.1 Stored Energy of Cold Work, 216 8.2 The Relationship of Free Energy to Strain Energy, 217 8.3 The Release of Stored Energy, 218 8.4 Recovery, 220 8.5 Recovery in Single Crystals, 221 8.6 Polygonization, 223 8.7 Dislocation Movements in Polygonization, 226 8.8 Recovery Processes at High and Low Temperatures, 229 8.9 Recrystallization, 230 8.10 The Effect of Time and Temperature on Recrystallization, 230 8.11 Recrystallization Temperature, 232 8.12 The Effect of Strain on Recrystallization, 233 8.13 The Rate of Nucleation and the Rate of Nucleus Growth, 234 8.14 Formation of Nuclei, 235 8.15 Driving Force for Recrystallization, 237 8.16 The Recrystallized Grain Size 237 8.17 Other Variables in Recrystallization, 239 8.18 Purity of the Metal, 239 8.19 Initial Grain Size, 240 8.20 Grain Growth, 240 8.21, Geometrical Coalescence, 243 8.22 Three-Dimensional Changes in Grain Geometry, 244 8.23 The Grain Growth Law, 245 8.24 Impurity Atoms in Solid Solution, 249 8.25 Impurities in the Form of Inclusions, 250 8.26 The Free-Surface Effects, 253 8.27 The Limiting Grain Size, 254 8.28 Preferred Orientation, 256 8.29 Secondary Recrystallization, 256 8.30 Strain-Induced Boundary Migration 257 Problems, 258 References, 259	

<b>CHAPTER 9</b>	<b>SOLID SOLUTIONS</b>	<b>261</b>
	9.1 Solid Solutions, 261 9.2 Intermediate Phases, 261 9.3 Interstitial Solid Solutions, 262 9.4 Solubility of Carbon in Body-Centered Cubic Iron, 263 9.5 Substitutional Solid Solutions and the Hume-Rothery Rules, 267 9.6 Interaction of Dislocations and Solute Atoms, 267 9.7 Dislocation Atmospheres, 268 9.8 The Formation of a Dislocation Atmosphere, 269 9.9 The Evaluation of $A$ , 270 9.10 The Drag of Atmospheres on Moving Dislocations, 271 9.11 The Sharp Yield Point and Lüders Bands, 273 9.12 The Theory of the Sharp Yield Point, 275 9.13 Strain Aging, 276 9.14 The Cottrell-Bilby Theory of Strain Aging, 277 9.15 Dynamic Strain Aging 282 Problems, 285 References, 286	

<b>CHAPTER 10</b>	<b>PHASES</b>	<b>287</b>
	10.1 Basic Definitions, 287 10.2 The Physical Nature of Phase Mixtures, 289 10.3 Thermodynamics of Solutions, 289 10.4 Equilibrium Between Two Phases, 292 10.5 The Number of Phases in an Alloy System, 293 10.6 Two-Component Systems Containing Two Phases, 303 10.7 Graphical Determinations of Partial-Molal Free Energies, 304 10.8 Two-Component Systems with Three Phases in Equilibrium, 306 10.9 The Phase Rule, 307 10.10 Ternary Systems, 309 Problems, 310 References, 311	

<b>CHAPTER 11</b>	<b>BINARY PHASE DIAGRAMS</b>	<b>312</b>
	11.1 Phase Diagrams, 312 11.2 Isomorphous Alloy Systems, 312 11.3 The Lever Rule, 314 11.4 Equilibrium Heating or Cooling of an Isomorphous Alloy, 317 11.5 The Isomorphous Alloy System from the Point of View of Free Energy, 319 11.6 Maxima and Minima, 320 11.7 Superlattices, 322 11.8 Miscibility Gaps, 326 11.9 Eutectic Systems, 328 11.10 The Microstructures of Eutectic Systems, 329 11.11 The Peritectic Transformation, 334 11.12 Monotectics, 337 11.13 Other Three-Phase Reactions, 338 11.14 Intermediate Phases, 339 11.15 The Copper-Zinc Phase Diagram, 341 11.16 Ternary Phase Diagrams, 343 Problems, 346 References, 347	
<b>CHAPTER 12</b>	<b>DIFFUSION IN SUBSTITUTIONAL SOLID SOLUTIONS</b>	<b>348</b>
	12.1 Diffusion in an Ideal Solution, 348 12.2 The Kirkendall Effect, 352 12.3 Pore Formation, 355 12.4 Darken's Equations, 357 12.5 Fick's Second Law, 360 12.6 The Matano Method, 363 12.7 Determination of the Intrinsic Diffusivities, 366 12.8 Self-Diffusion in Pure Metals, 368 12.9 Temperature Dependence of the Diffusion Coefficient, 370 12.10 Chemical Diffusion at Low-Solute Concentration, 372 12.11 The Study of Chemical Diffusion Using Radioactive Tracers, 374 12.12 Diffusion Along Grain Boundaries and Free Surfaces, 377 12.13 Fick's First Law in Terms of a Mobility and an Effective Force, 380 12.14 Diffusion in Non-Isomorphic Alloy Systems, 382 Problems, 386 References, 388	
<b>CHAPTER 13</b>	<b>INTERSTITIAL DIFFUSION</b>	<b>389</b>
	13.1 Measurement of Interstitial Diffusivities, 389 13.2 The Snoek Effect, 391 13.3 Experimental Determination of the Relaxation Time, 398 13.4 Experimental Data, 405 13.5 Anelastic Measurements at Constant Strain, 405 Problems, 406 References, 407	
<b>CHAPTER 14</b>	<b>SOLIDIFICATION OF METALS</b>	<b>408</b>
	14.1 The Liquid Phase, 408 14.2 Nucleation, 411 14.3 Metallic Glasses, 413 14.4 Crystal Growth from the Liquid Phase, 420 14.5 The Heats of Fusion and Vaporization, 421 14.6 The Nature of the Liquid-Solid Interface, 423 14.7 Continuous Growth, 425 14.8 Lateral Growth, 427 14.9 Stable Interface Freezing, 428 14.10 Dendritic Growth in Pure Metals, 429 14.11 Freezing in Alloys with Planar Interface, 432 14.12 The Scheil Equation, 434 14.13 Dendritic Freezing in Alloys, 437 14.14 Freezing of Ingots, 439 14.15 The Grain Size of Castings, 443 14.16 Segregation, 443 14.17 Homogenization 445 14.18 Inverse Segregation, 450 14.19 Porosity, 450 14.20 Eutectic Freezing, 454 Problems, 459 References, 461	

<b>CHAPTER 15</b>	<b>NUCLEATION AND GROWTH KINETICS</b>	<b>463</b>
	15.1 Nucleation of a Liquid from the Vapor, 463 15.2 The Becker-Döring Theory, 471 15.3 Freezing, 473 15.4 Solid-State Reactions, 475 15.5 Heterogeneous Nucleation, 478 15.6 Growth Kinetics, 481 15.7 Diffusion Controlled Growth, 484 15.8 Interference of Growing Precipitate Particles, 488 15.9 Interface Controlled Growth, 488 15.10 Transformations That Occur on Heating, 492 15.11 Dissolution of a Precipitate, 493 Problems, 495 References, 497	
<b>CHAPTER 16</b>	<b>PRECIPITATION HARDENING</b>	<b>498</b>
	16.1 The Significance of the Solvus Curve, 499 16.2 The Solution Treatment, 500 16.3 The Aging Treatment, 500 16.4 Development of Precipitates, 503 16.5 Aging of Al-Cu Alloys at Temperatures Above 100°C (373 K), 506 16.6 Precipitation Sequences in Other Aluminum Alloys, 509 16.7 Homogeneous Versus Heterogeneous Nucleation of Precipitates, 511 16.8 Interphase Precipitation, 512 16.9 Theories of Hardening, 515 16.10 Additional Factors in Precipitation Hardening, 516 Problems, 518 References, 519	
<b>CHAPTER 17</b>	<b>DEFORMATION TWINNING AND MARTENSITE REACTIONS</b>	<b>521</b>
	17.1 Deformation Twinning, 521 17.2 Formal Crystallographic Theory of Twinning, 524 17.3 Twin Boundaries, 530 17.4 Twin Growth, 531 17.5 Accommodation of the Twinning Shear, 533 17.6 The Significance of Twinning in Plastic Deformation, 534 17.7 The Effect of Twinning on Face-Centered Cubic Stress-Strain Curves, 535 17.8 Martensite, 537 17.9 The Bain Distortion, 538 17.10 The Martensite Transformation in an Indium-Thallium Alloy, 540 17.11 Reversibility of the Martensite Transformation, 541 17.12 Athermal Transformation, 541 17.13 Phenomenological Crystallographic Theory of Martensite Formation, 542 17.14 Irrational Nature of the Habit Plane, 548 17.15 The Iron-Nickel Martensitic Transformation, 549 17.16 Isothermal Formation of Martensite, 551 17.17 Stabilization, 551 17.18 Nucleation of Martensite Plates, 552 17.19 Growth of Martensite Plates, 553 17.20 The Effect of Stress, 553 17.21 The Effect of Plastic Deformation, 554 17.22 Thermoelastic Martensite Transformations, 554 17.23 Elastic Deformation of Thermoelastic Alloys, 556 17.24 Stress-Induced Martensite (SIM), 556 17.25 The Shape-Memory Effect, 557 Problems, 559 References, 560	
<b>CHAPTER 18</b>	<b>THE IRON-CARBON ALLOY SYSTEM</b>	<b>562</b>
	18.1 The Iron-Carbon Diagram, 562 18.2 The Proeutectoid Transformations of Austenite, 565 18.3 The Transformation of Austenite to Pearlite, 566 18.4 The Growth of Pearlite, 572 18.5 The Effect of Temperature on the Pearlite	

Transformation, 573 **18.6** Forced-Velocity Growth of Pearlite, 575 **18.7** The Effects of Alloying Elements on the Growth of Pearlite, 578 **18.8** The Rate of Nucleation of Pearlite, 581 **18.9** Time-Temperature-Transformation Curves, 583 **18.10** The Bainite Reaction, 584 **18.11** The Complete T-T-T Diagram of an Eutectoid Steel, 591 **18.12** Slowly Cooled Hypoeutectoid Steels, 593 **18.13** Slowly Cooled Hypereutectoid Steels, 595 **18.14** Isothermal Transformation Diagrams for Noneutectoid Steels, 597 Problems, 600 References, 602

---

**CHAPTER 19 THE HARDENING OF STEEL 603**

**19.1** Continuous Cooling Transformations (CCT), 603 **19.2** Hardenability, 606 **19.3** The Variables that Determine the Hardenability of a Steel, 614 **19.4** Austenitic Grain Size, 614 **19.5** The Effect of Austenitic Grain Size on Hardenability, 615 **19.6** The Influence of Carbon Content on Hardenability, 615 **19.7** The Influence of Alloying Elements on Hardenability, 616 **19.8** The Significance of Hardenability, 621 **19.9** The Martensite Transformation in Steel, 622 **19.10** The Hardness of Iron-Carbon Martensite, 627 **19.11** Dimensional Changes Associated with Transformation of Martensite, 631 **19.12** Quench Cracks, 632 **19.13** Tempering, 633 **19.14** Tempering of a Low-Carbon Steel, 639 **19.15** Spheroidized Cementite, 641 **19.16** The Effect of Tempering on Physical Properties, 643 **19.17** The Interrelation Between Time and Temperature in Tempering, 646 **19.18** Secondary Hardening, 646 Problems, 647 References, 649

---

**CHAPTER 20 SELECTED NONFERROUS ALLOY SYSTEMS 651**

**20.1** Commercially Pure Copper, 651 **20.2** Copper Alloys, 654 **20.3** Copper Beryllium, 658 **20.4** Other Copper Alloys, 659 **20.5** Aluminum Alloys, 659 **20.6** Aluminum-Lithium Alloys, 660 **20.7** Titanium Alloys, 668 **20.8** Classification of Titanium Alloys, 670 **20.9** The Alpha Alloys, 670 **20.10** The Beta Alloys, 676 **20.11** The Alpha-Beta Alloys, 677 **20.12** Superalloys, 679 **20.13** Creep Strength, 680 Problems, 683 References, 684

---

**CHAPTER 21 FAILURE OF METALS 686**

**21.1** Failure by Easy Glide, 686 **21.2** Rupture by Necking (Multiple Glide), 688 **21.3** The Effect of Twinning, 689 **21.4** Cleavage, 690 **21.5** The Nucleation of Cleavage Cracks, 691 **21.6** Propagation of Cleavage Cracks, 693 **21.7** The Effect of Grain Boundaries, 696 **21.8** The Effect of the State of Stress, 698 **21.9** Ductile Fractures, 700 **21.10** Intercrystalline Brittle Fracture, 705 **21.11** Blue Brittleness, 705 **21.12** Fatigue Failures, 706 **21.13** The Macroscopic Character of Fatigue Failure, 706 **21.14** The Rotating-Beam Fatigue Test, 708 **21.15** Alternating Stress Parameters, 710 **21.16** The Microscopic Aspects of Fatigue Failure, 713 **21.17** Fatigue Crack Growth, 717 **21.18** The Effect of Nonmetallic Inclusions, 720 **21.19** The Effect of Steel

Microstructure on Fatigue, 721 **21.20** Low-Cycle Fatigue, 721 **21.21** The Coffin-Manson Equation, 726 **21.22** Certain Practical Aspects of Fatigue, 727 Problems, 728 References, 729

---

**APPENDICES**
**731**

- A**     **ANGLES BETWEEN CRYSTALLOGRAPHIC PLANES IN THE CUBIC SYSTEM\* (IN DEGREES), 731**
- B**     **ANGLES BETWEEN CRYSTALLOGRAPHIC PLANES FOR HEXAGONAL ELEMENTS\*, 733**
- C**     **INDICES OF THE REFLECTING PLANES FOR CUBIC STRUCTURES, 734**
- D**     **CONVERSION FACTORS AND CONSTANTS, 734**
- E**     **TWINNING ELEMENTS OF SEVERAL OF THE MORE IMPORTANT TWINNING MODES, 735**
- F**     **SELECTED VALUES OF INTRINSIC STACKING-FAULT ENERGY  $\gamma_p$ , TWIN-BOUNDARY ENERGY  $\gamma_T$ , GRAIN-BOUNDARY ENERGY  $\gamma_G$ , AND CRYSTAL-VAPOR SURFACE ENERGY  $\gamma$  FOR VARIOUS MATERIALS IN ERGS/CM<sup>2</sup>\*, 735**

**LIST OF IMPORTANT SYMBOLS****737****LIST OF GREEK LETTER SYMBOLS****739****INDEX****740**



# The Structure of Metals

The most important aspect of any engineering material is its structure, because its properties are closely related to this feature. To be successful, a materials engineer must have a good understanding of this relationship between structure and properties. By way of illustration, wood is a very easy material in which to see the close interaction between structure and properties. A typical structural wood, such as southern yellow pine, is essentially an array of long hollow cells or fibers. These fibers, which are formed largely from cellulose, are aligned with the grain of the wood and are cemented together by another weaker organic material called lignin. The structure of wood is thus roughly analogous to that of a bundle of drinking straws. It can be split easily along its grain; that is, parallel to the cells. Wood is also much stronger in compression (or tension) parallel to its grain than it is in compression (or tension) perpendicular to the grain. It makes excellent columns and beams, but it is not really suitable for tension members required to carry large loads, because the low resistance of wood to shear parallel to its grain makes it difficult to attach end fastenings that will not pull out. As a result, wooden bridges and other large wooden structures are often constructed containing steel tie rods to support the tensile loads.

## 1.1 THE STRUCTURE OF METALS

The structure in metals is of similar importance to that in wood, although often in a more subtle manner. Metals are usually crystalline when in the solid form. While very large single crystals can be prepared, the normal metallic object consists of an aggregate of many very small crystals. Metals are therefore *polycrystalline*. The crystals in these materials are normally referred to as its grains. Because of their very small sizes, an optical microscope, operating at magnifications between about 100 and 1000 times, is usually used to examine the structural features associated with the grains in a metal. Structures requiring this range of magnification for their examination fall into the class known as *microstructures*. Occasionally, metallic objects, such as castings, may have very large crystals that are discernible to the naked eye or are easily resolved under a low-power microscope. Structure in this category is called *macrostructure*. On the other hand, there are materials whose grains or sizes are much finer and in the nanoscale range. These microstructures are commonly referred to as *nanosstructures*, with scales on the order of one billionth of a meter. It should be noted that nanoscale features can be in one dimension, as in nanosurfaces or nanofilms; in two dimensions, as in nanotubes or whiskers; or in three dimensions, as in nanoparticles. Nanoprecipitates such as Guinier and Preston (GP) Zones have been used for decades for precipitation hardening of aluminum alloys, as discussed in Chapter 16. Finally, there is the basic structure inside the grains themselves: that is, the atomic arrangements inside the crystals. This form of structure is logically called the *crystal structure*.

Of the various forms of structure, microstructure (that visible under the optical microscope) has been historically of the greatest use and interest to the metallurgist. Because the metallurgical microscope is normally operated at magnifications where its

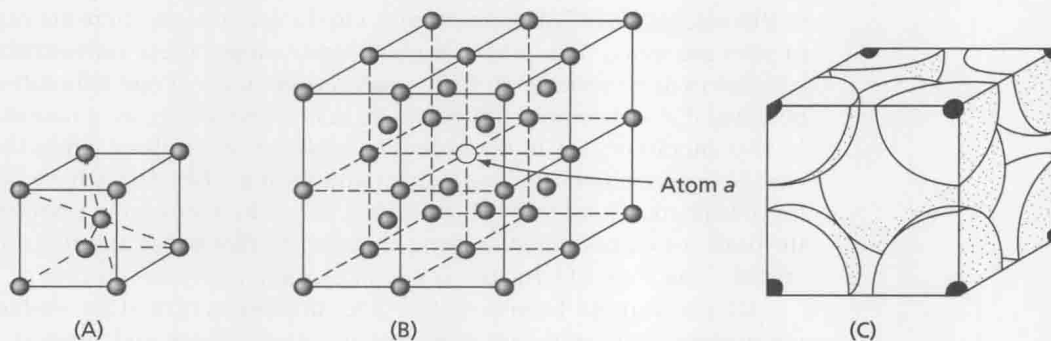
depth of field is extremely shallow, the metallic surface to be observed must be very flat. At the same time, it must reveal accurately the nature of the structure inside the metal. One is therefore presented with the problem of preparing a very smooth flat and undistorted surface, which is by no means an easy task. The procedures required to obtain the desired goal fall under the general heading of metallographic specimen preparation. Detailed description of metallographic sample preparation techniques and examples of microstructures can be found in Reference 1.

A *crystal* is defined as an orderly array of atoms in space. There are many different types of crystal structures, some of which are quite complicated. Fortunately, most metals crystallize in one of three relatively simple structures: the face-centered cubic, the body-centered cubic, and the close-packed hexagonal.

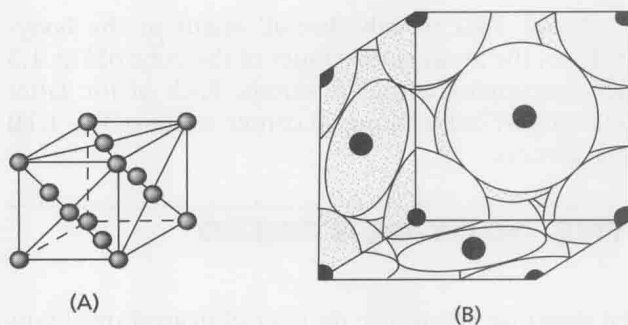
## 1.2 UNIT CELLS

The *unit cell* of a crystal structure is the smallest group of atoms possessing the symmetry of the crystal which, when repeated in all directions, will develop the crystal lattice. Figure 1.1A shows the unit cell of the body-centered cubic lattice. It is evident that its name is derived from the shape of the unit cell. Eight unit cells are combined in Fig. 1.1B in order to show how the unit cell fits into the complete lattice. Note that atom *a* of Fig. 1.1B does not belong uniquely to one unit cell, but is a part of all eight unit cells that surround it. Therefore, it can be said that only one-eighth of this corner atom belongs to any one-unit cell. This fact may be used to compute the number of atoms per unit cell in a body-centered cubic crystal. Even a small crystal will contain billions of unit cells, and the cells in the interior of the crystal must greatly exceed in number those lying on the surface. Therefore, surface cells may be neglected in our computations. In the interior of a crystal, each corner atom of a unit cell is equivalent to atom *a* of Fig. 1.1B and contributes one-eighth of an atom to a unit cell. In addition, each cell also possesses an atom located at its center that is not shared with other unit cells. The body-centered cubic lattice thus has two atoms per unit cell; one contributed by the corner atoms, and one located at the center of the cell, as shown in Fig. 1.1C.

The unit cell of the face-centered cubic lattice is shown in Fig. 1.2. In this case, the unit cell has an atom in the center of each face. The number of atoms per unit cell in the face-centered cubic lattice can be computed in the same manner as in the body-centered



**FIG. 1.1** (A) Body-centered cubic unit cell. (B) Eight unit cells of the body centered cubic lattice. (C) Cut view of a unit cell



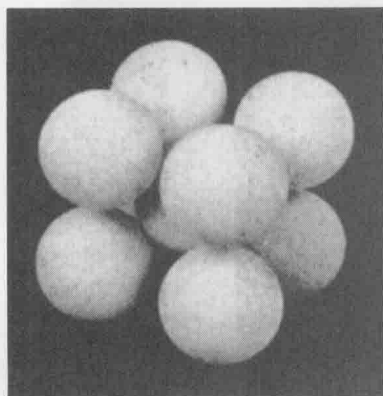
**FIG. 1.2** (A) Face-centered cubic unit cell.  
(B) Cut view of a unit cell

cubic lattice. The eight corner atoms again contribute one atom to the cell, as shown in Fig. 1.2B. There are also six face-centered atoms to be considered, each a part of two unit cells. These contribute six times one-half an atom, or three atoms. The face-centered cubic lattice has a total of four atoms per unit cell, or twice as many as the body-centered cubic lattice.

### 1.3 THE BODY-CENTERED CUBIC STRUCTURE (BCC)

It is frequently convenient to consider metal crystals as structures formed by stacking together hard spheres. This leads to the so-called *hard-ball model* of a crystalline lattice, where the radius of the spheres is taken as half the distance between the centers of the most closely spaced atoms.

Figure 1.3 shows the hard-ball model of the body-centered cubic (bcc) unit cell. A study of the figure shows that the atom at the center of the cube is colinear with each corner atom; that is, the atoms connecting diagonally opposite corners of the cube form straight lines, each atom touching the next in sequence. These linear arrays do not end at the corners of the unit cell, but continue on through the crystal much like a row of beads strung on a wire (see Fig. 1.1B). These four cube diagonals constitute the close-packed directions of the body-centered cubic crystal, directions that run continuously through the lattice on which the atoms are as closely spaced as possible.



**FIG. 1.3** Hard-ball model of the body-centered cubic unit cell