



RICHARD S STEIN

Goessmann Professor of Chemistry, Emeritus University of Massachusetts, Amherst

Joseph Powers

United Technologies Corporation, retired

Published by

Imperial College Press 57 Shelton Street Covent Garden London WC2H 9HE

Distributed by

World Scientific Publishing Co. Pte. Ltd. 5 Toh Tuck Link, Singapore 596224

USA office: 27 Warren Street, Suite 401-402, Hackensack, NJ 07601 UK office: 57 Shelton Street, Covent Garden, London WC2H 9HE

British Library Cataloguing-in-Publication Data

A catalogue record for this book is available from the British Library.

TOPICS IN POLYMER PHYSICS

Copyright © 2006 by Imperial College Press

All rights reserved. This book, or parts thereof, may not be reproduced in any form or by any means, electronic or mechanical, including photocopying, recording or any information storage and retrieval system now known or to be invented, without written permission from the Publisher.

For photocopying of material in this volume, please pay a copying fee through the Copyright Clearance Center, Inc., 222 Rosewood Drive, Danvers, MA 01923, USA. In this case permission to photocopy is not required from the publisher.

ISBN 1-86094-411-6 ISBN 1-86094-412-4 (pbk)

TO OUR WIVES

JUDY BALISE STEIN MARY GRIFFIN POWERS

Preface

This book can serve as an introduction to students interested in learning the techniques used in developing mathematical models of physical phenomenon; or it can furnish the background information to the experienced professional desiring to broaden his/her knowledge of polymers.

The senior author presented material in this book to students interested in learning the fundamental mathematics underlying many areas of polymer physics and in lectures to audiences with varying backgrounds in polymer physics.

The material in this book should prove helpful to readers who have knowledge of introductory mathematics, chemistry and physics.

The text emphasizes the derivation of many equations used in Polymer Physics. The assumptions used in modeling, and in making the mathematical apparatus solvable in closed form, are presented in detail. Too many times, the basic equations are presented in final form in journal articles and books from either lack of space or the assumption that the derivation is widely disseminated and does not require repetition.

The fundamentals of any discipline have to be constantly tested against new findings. This book presents the assumptions and simplifications of the fundamentals of many areas of Polymer Physics so that the testing process can be expedited.

The authors have discussed this material with many colleagues and in return received many pertinent suggestions for improvement. These include Philip Wilson, Mohan Srinivasaro, Hiromichi Kawai, Shigeharu Onogi, Garth Wilkes, Takeji Hashimoto and Marion Rhodes, James J. Burke and many others who attended courses or collaborated with Professor Stein in research projects.

The improvements belong to our colleagues. The residual factual errors, typos and other problems belong to the authors.

For supplementary material, corrections and communications with us, please visit http://web.mac.com/rsstein1/iWeb

Contents

Dedication	V
Preface	vii
Chapter 1	1
INTRODUCTION	1
1.1. Background	1
1.2. Linear Chain Molecules	4
1.2.1. Structure	4
1.2.2. Synthesis	
1.2.3. Molecular Weight	.10
1.3. Network Molecules	.15
1.3.1. Structure	.15
1.3.2. Synthesis	.16
1.3.3. Molecular Weight	.16
1.4. Rubber	.17
1.4.1. Structure	.17
1.4.2. Synthesis	.21
1.4.3. Molecular Weight	.21
1.5. Multicomponent Systems	.21
1.5.1. Copolymers	.21
1.5.2. Polyblends	.25
1.5.3. Composites	.25
1.5.4. Ceramers	.26
References	.27
Appendix 1A - Derivation of the Most Probable Distribution	128
Reference	
Chapter 2	
STATISTICS OF CHAIN CONFORMATIONS	.31
2.1. Introduction	.31
2.2. Small Molecules	.31
2.3. Larger Molecules, Statistical Variation of Molecular	
Conformations	
2.4. Statistical Segment Model	.41
2.5. Generalization to High Polymers	
2.6. Polymer Chains Containing Two Kinds of Atoms	.45

2.7. Model Chains with Restricted Rotation and No Inter-	action
Among the ϕ 's	
2.8. Rotational Isomeric State (RIS) Approximation	49
2.9. Chains with Interactions between ϕ Values of Neigh	
Monomers	51
2.10. Asymmetric Barriers	53
2.11. Comparison with Experiment	
2.12. Chain End-to-End Distribution Functions	58
2.12.1 One Dimensional Case	58
2.12.2. Extension to a Three Dimensional Chain	61
2.12.3. Extension to Non-Gaussian Case	64
References	71
Appendix 2A - Statistical Mechanics	73
References	80
Statistical Thermodynamics of an Ideal Monatomic Gas	81
Appendix 2B - Vector Analysis	
2B.1. Vector Addition	
2.B.2 Scalar Product	84
Reference	
Appendix 2C - Radius of Gyration	86
Appendix 2D. Evaluation of $(\mathbf{a}_{m+n} \cdot \mathbf{a}_{m})$	92
Appendix 2E. Restricted Rotation with Symmetrical Bar	rier.99
Appendix 2F. Rotational Isomeric State (RIS) Approxim	ation
	103
2F.1. Introduction	
2F.2. The Conformational Partition Function	104
2E.3. Mean Square Distance	106
References	107
Appendix 2G. Random Walk of Gaussian Chains	109
References	113
Appendix 2H. Radius of Gyration, Size and Shape	114
Definition of $\left\langle \mathbf{r}_{\mathbf{q}}^{2}\right\rangle$	114
$\left\langle \mathbf{r_q^2} \right angle$ for a Rigid Rod	115
$\langle \mathbf{r_q}^2 \rangle$ for a Sphere of Radius R	116
1 1	

CONTENTS

Determination of $\langle {\bf r_q^2} \rangle$ from Scattering	117
Chapter 3	121
THERMODYNAMICS	121
3.1. Introduction	
3.2 Thermodynamics of Elasticity	121
3.3. Force on a Chain in the Presence of Energy Contribut	tions
to Elasticity	129
3.4. Solution Thermodynamics	134
3.4.1. Flory-Huggins Theory	134
3.4.2. Calculation of P ₂	138
3.3.3. Entropy of Mixing	140
3.5. Polymer Miscibility	
3.6. Spinodal Decomposition	
3.7. Heat Capacity	153
3.8. Thermodynamics of Crystallization	
3.8.1 Polymer Structure – Melting Point	155
3.8.2. Effect of Diluent and Impurities	
3.8.3. Crystallization Induced by Stretching	
References	167
Appendix 3A Classical Thermodynamics	169
References	175
Appendix 3B. Heat of Mixing	176
Appendix 3C Osmotic Pressure	181
Chapter 4	185
OPTICS	185
4.1. Introduction	185
4.2. Nature of Electromagnetic Waves	186
4.3. Refraction	187
4.4. Scattering	191
4.4.1. Intensity of Light Scattering for an Isolated Ato	m or
Molecule	194
4.4.2. Effect of the Polarization of Light	195
4.4.3. The Scattering Intensity for a Collection of	
Scattering Objects	196
4.5 Diffraction	205

4.5.1.The Reciprocal Lattice	.205
4.5.2. Interpretation of the Vector Bragg Equation	
4.5.3. The Distance Between Crystal Planes	.209
4.5.4. The Diffraction Phenomenon in One Dimension.	
4.6. Absorption	.215
4.7. Fluorescence	.216
4.8. Birefringence	.218
4.9. Scattering from Inhomogeneous Media	.222
4.10. Quasi-elastic Light Scattering	.228
4.11. Variation of Scattering with Electric Fields	
4.12. Non-Linear Optics	
4.13. Piezo-Electric Materials	.232
4.14. Kerr Effect	.233
References	234
Appendix 4A Depolarization of Scattering	236
Appendix 4B Scattering from a Collection of Molecules	.239
Appendix 4D The Magnitude of the Reciprocal Vector H	244
Appendix 4E Orientation Fluctuations	246
Appendix 4F Scattering from Concentration Fluctuations	
Evaluation of $(\partial \alpha/\partial c)$	252
Evaluation of $\left[-\left(\partial \mu_1/\partial c\right)\right]$	252
Molecular Weight from Light Scattering	253
Dealing with Large Molecules	255
Extensions to Other Kinds of Radiation	256
Extensions to Concentrated Solution and Bulk Polymer	rs
Effect of Anisotropy and Orientation	
Appendix 4G Why Concentration Fluctuations Relate t	
Molecular Weight	
How to Measure the Size of an Elephant	
ANOTHER WAY OF LOOKING AT IT	
Chapter 5	
ELECTRICAL PROPERTIES	
5.1 Introduction	263
5.2. Dielectrics	264

CONTENTS

5.2.1. Dielectric Constant	264
5.2.2. Orientation of permanent dipoles	269
5.2.3. Dielectric Loss	
5.3. Piezo- and Pyroelectric Effects	272
5.4. Piezo-Electric Coefficient	274
References	276
Appendix 5A. Lines of Flux	
5A.1. Electrical Field Strength	278
5A.2. Electric displacement and Flux Density	278
5A.4. The Electrostatic Potential (Voltage)	280
5A.5. The Field between Parallel Plates	280
Appendix 5B. Lorentz Calculation, Internal Field Correction	on
	282
Chapter 6	285
SPECTROSCOPY	
6.1. Introduction	285
6.2. General Background	286
6.3. Infrared	288
6.4. Raman	295
6.5. Ultraviolet and Visible	299
6.6. Nuclear Magnetic Resonance	300
6.7. Neutron Inelastic Scattering (NIS)	301
References	
Appendix 6A Fourier Transform Infrared (FTIR)	305
References	307
Appendix 6B. Normal Coordinate Analysis	308
References	312
Appendix 6C. Spectrographic Notation	313
Chapter 7	314
THE RUBBERY STATE	314
7.1. Introduction	314
7.2. Force - Extension Relation for Rubbers	316
7.2.1. Simple Case	316
7.2.2. Consideration of Network Crosslinks	318
7.3. Affine Transformation	
7.4. Uniaxial Stretching at Constant Volume	329

7.5. Biaxial Stretching at Constant Volume	331
7.6. Application to the Inflation of a Balloon	333
7.7. Network Defects -The Relationship between N _c and	$\nu.334$
7.8. Effect of Swelling on an Isotropic Network	336
7.9. Elastic Properties of Swollen Rubber	
7.10. Elasticity of a Sample That is Swollen When Cross	
	341
7.11. Elasticity of Rubbers at Small Extensions	
7.12. Guth-Smallwood Equations	347
References	
Appendix 7A – Evaluation of Equation 7.30	350
CHAPTER 8	
THE CRYSTALLINE STATE	352
8.1. Introduction	352
8.2. Evidence for Crystallinity	354
8.2.1. X-Ray Diffraction	354
8.2.2. Electron Microscopical Observations	358
8.2.3. Optical Microscopy	359
8.2.4. Thermodynamic Transitions	360
8.3. Determination of Degree of Crystalline	361
8.3.1. Density Measurements	361
8.3.2. X-Ray Diffraction	364
8.3.3. Infrared (IR) and Raman	
8.3.4. Wide Line Nuclear Magnetic Resonance	366
8.3.5. Thermal Measurements	368
8.4. Morphology	369
8.4.1. Electron Microscope	
i) The observed pyramidal structure	369
ii) Moiré Patterns	370
i.) Density	370
ii) nmr	371
iii) X-ray diffraction	
iv) Kinetics	
8.4.2. Optical Microscopy	372
8.4.3. Liquid Crystals	
8.5 Mechanisms of Crystallization	375

CONTENTS

8.5.1. Nucleation and Growth	375
8.6. Kinetics of Crystallization	376
8.6.1. Temperature Dependence of the Nucleation Rate.	376
8.6.2. Nucleation and Growth	379
8.6.3. Experimental Determination of n and k	384
i. Slope-Intercept Method	384
ii. Half-life Method	385
8.6.4. Temperature Dependence of the Rate of	
Homogeneous Crystallization	386
References	.387
Glossary of Symbols Arranged by Chapters	.389
Chapter 1	.389
Chapter 2	
Chapter 3	
Chapter 4	
Chapter 5	
Chapter 7	
Chapter 8	
Index	

Chapter 1

INTRODUCTION

1.1. Background

The concept of long chain molecules in which the atoms forming the backbone are bonded by strong primary valence forces, usually covalent in nature, is the foundation of polymer science. Until relatively recently, chemistry, which is the study of molecules in all their ramifications, emphasized the study of small molecules. These could be readily purified to yield materials with constant composition and well characterized properties. Naturally occurring polymers, such as natural rubber or cellulose, did not fit into this framework of small well-characterized molecules [1]. Many investigators considered that the naturally occurring polymers constituted a fourth state of matter, essentially colloidal in nature, because these materials did not seem to obey the laws derived for gases, liquids, or solids as these laws were then understood. The assertion that, since rubber and cellulose were the products of living organisms, a vital principle, not amenable to physico-chemical laws, was involved reinforced this viewpoint.

Based on chemical evidence, Staudinger [2], in the early 1920s, was the first investigator to strongly advocate long, linear chain structures for polystyrene, polyoxymethylene, and natural rubber. During this time span, x-ray diffraction developed as a tool for determining the structure of molecules. The diffraction photographs of natural rubber and cellulose taken by Meyer and Mark were interpreted as showing that these polymers did have long chain structures [3]. Carothers furnished a key argument, based on purely chemical reasoning, in favor of the view that polymers, in the main, were composed of long linear chains. By the late

1920s, organic chemists had accumulated a large store of knowledge on the reaction conditions, products, yields and structures of many small monomeric molecules involved in organic chemical reactions. For example, the reaction between an amine and an organic acid was known to produce an amide with the elimination of a molecule of water:

$$R_1NH_2 + HOOCR_2 \rightarrow R_1NHCOR_2 + H_2O$$

 $(R_1, R_2 \text{ being typically methyl}, \text{ ethyl or other aliphatic radicals}).$ Carothers reasoned that, if both molecules were diffunctional (i.e., two amine groups on R_1 and two organic acid groups on R_2), sites would be available for further reactions:

$$H_2NR_1NH_2 + HOOCR_2COOH \rightarrow H_2NR_1NHCOR_2COOH + H_2O$$

The reaction would continue until the starting materials (diffunctional amines and organic acids) were exhausted. Thus, a long chain polymer structure could be synthesized using a well known and well understood organic chemical reaction. In a classic series of investigations [4], Carothers and a small group of co-workers were able to demonstrate that this and similar chemical reactions produced long linear chain molecules. As a point of interest, Carothers produced nylon 66 by using C₆H₁₂ (hexamethylene) for R₁ and C₄H₈ for R₂. Carothers [4] coined the term condensation polymers for the long chain molecules produced by these reactions because the elimination of small molecules such as water condensed the length of the polymer repeat unit compared to that of the starting molecules. Conversely, Carothers called polymers such as polystyrene and polyoxymethylene addition polymers because the monomer units add through the opening of double bonds. Thus, these latter polymers added monomers during formation with no elimination of small molecules.

Many of these linear chain polymers have the advantage for characterization that they are soluble in organic solvents. This has aided greatly, as will be shown later, in the analysis of chain structures and reaction mechanisms and the determination of molecular weight. Most linear polymers, both condensation and addition, also reversibly soften and flow on heating and conversely harden and become rigid on cooling. These materials are sometimes called thermoplastics because they flow

INTRODUCTION

at sufficiently high temperatures. This thermal characteristic is used to advantage in reprocessing these polymers. The thermoplastics did not become commercially available, with a few exceptions such as cellulose acetate, polystyrene, poly(methyl methacrylate), until after World War II. A second class of polymeric materials (thermosets) had been introduced earlier (ca. 1910) starting with the phenol-formaldehyde polymers developed by Leo Baekeland [5]. Thermosets are composed of non-linear polymer chains (based on a functionality of 3 or more) that combine chemically to form three dimensional network polymers. They are soluble and fusible only up through the intermediate stages of polymerization. Once polymerization is complete, the thermosets form hard infusible insoluble structures that soften on heating over a temperature range. The average temperature for the range is called the glass temperature. On heating to higher temperatures, thermosets decompose because of their network structure. The lack of solubility and general intractability rendered thermosets difficult to study from a fundamental standpoint.

Elastomers or rubbers represent an intermediate stage in terms of functionality between thermoplastics and thermosets. Elastomeric behavior in polymers originates from a special type of chain structure. This point is discussed more fully in Chapter 7. Elastomers to be useful articles of commerce require a controlled number of crosslinks (chemical bonds between neighboring chains). Natural rubber (and its progeny, the synthetic rubbers, spawned during and after World War II) elongates readily to several times its original length on the application of a small force and readily retracts with release of the applied force. But, unless it is crosslinked, rubber will tend to flow on being held at high elongations and gradually lose its ability to retract. Goodyear [6] found in the 1830s that, upon adding sulfur to natural rubber latex and heating, the coagulated latex changed from a flowable gum to a retractive elastic material. He also observed that the hardness of natural rubber ranged from a soft crepe rubber (essentially thermoplastic) on small sulfur additions to hard infusible rubber (or thermoset) with large sulfur additions.

Figure 1.1a. Linear Polyethylene.

Figure 1.1b. Branched Polyethylene.

1.2. Linear Chain Molecules

1.2.1. Structure

Polyethylene, the prototype or model chain for linear addition polymers, is composed of ethylene monomer units linked by covalent bonds to form long chains (Figure 1.1a.) But, this linear chain was only produced in the mid 50's by the Ziegler–Natta catalysts. Before this, the Fawcett process that required high pressure and temperature produced a polyethylene chain that contained many small side chains or branches attached by covalent bonds to the main chain (Figure 1.1b). Nuclear

4