

THE PHYSICS OF RUBBER ELASTICITY

BY L. R. G. TRELOAR

THIRD EDITION

Oxford University Press, Ely House, London W.1
GLASGOW NEW YORK TORONTO MELBOURNE WELLINGTON
CAPE TOWN IBADAN NAIROBI DAR ES SALAAM LUSAKA ADDIS ABABA
DELHI BOMBAY CALCUTTA MADRAS KARACHI LAHORE DACCA
KUALA LUMPUR SINGAPORE HONG KONG TOKYO

ISBN 0 19 851355 0

©OXFORD UNIVERSITY PRESS 1975

All rights reserved. No part of this publication may be reproduced, stored in a retrieval system, or transmitted, in any form or by any means, electronic, mechanical, photocopying, recording or otherwise, without the prior permission of Oxford University Press

MONOGRAPHS ON THE PHYSICS AND CHEMISTRY OF MATERIALS

General Editors
C. E. H. BAWN, H. FRÖHLICH,
P. B. HIRSCH, N. F. MOTT

PREFACE TO THE THIRD EDITION

THE preparation of the Third Edition of this book has presented problems which were not encountered either with the First or with the Second Edition. The expansion of the subject during the last 16 years has involved a problem in the selection and arrangement of material to which there is no completely satisfactory solution. As a guiding principle I have assumed that the primary objective should be to provide a logical and reasonably detailed presentation of the main developments in the field of the equilibrium elastic properties of rubber (including the photoelastic and swelling properties), together with the associated theoretical background. In consequence it has been necessary to eliminate the last two chapters of the Second Edition, dealing respectively with stress-relaxation and flow and dynamic properties. The two chapters relating to crystallization have also been removed, though some references to this subject have been included in an enlarged Chapter 1. Despite its great inherent interest, particularly in relation to the historical development of the physics of rubber elasticity, the subject of crystallization in rubber is now seen to be incidental rather than fundamental to the main theme of this book, and its proper treatment would require an extensive discussion of crystallization in polymers other than rubber. A number of authoritative treatments of this wider subject are already in existence.

The main advances in more recent years have been in the thermodynamic analysis of rubber elasticity and in the essentially separate development of the phenomenological (i.e. non-molecular) approach to the subject. To take account of the former it has regrettably been necessary to divide the treatment of the thermodynamics into two parts, the first (elementary) being contained in Chapter 2, and the second (advanced) in the final chapter. The previous Chapter 8 (on phenomenological theory) has been expanded into three separate chapters (Chapters 10, 11, and 12), of which Chapter 11 contains essentially new material.

Inevitably these changes will to some extent reduce the attractiveness of the book for the student who wishes to acquire a broad knowledge of the whole range of physical phenomena associated with rubber. It can only be hoped that this loss will be more than

offset by the greater value of the book as a critical review of the subject of rubber elasticity in the more restricted sense, in which field no comparable work is readily available.

L.R.G.T.

Department of Polymer and Fibre Science, University of Manchester Institute of Science and Technology

PREFACE TO THE FIRST EDITION

It is sometimes considered unnecessary for those engaged in the practical development of industrial processes to concern themselves with the so-called theoretical aspects of their subject. On examination, it is usually found that exponents of this point of view are not entirely consistent, for in any type of work involving experimentation it is impossible to get along without some sort of theory, however limited or ad hoc it may be. My excuse for doing the work which I do (of which this book is one aspect) is that I always believe that if one is going to have a theory at all one may as well take some trouble to find the one which most nearly represents the known facts.

In the subject of rubber elasticity it is not easy to discover from the mass of literature, often of a rather mathematical character, what are the generally accepted theories. In this book I have therefore attempted to convey (in not too mathematical language) the fundamental concepts of the subject, and to present the whole in a more or less consistent form. In this task I have admittedly given expression to my own point of view, and I have drawn freely on the work of my associates at the British Rubber Producers' Research Association. I cannot hope to acknowledge the many who have helped me by the discussion of particular sections, but I should like to mention particularly Dr. G. Gee, Director of the B.R.P.R.A., who read and criticized the manuscript in detail, my colleague Mr. R. S. Rivlin, who gave me the benefit of his unpublished ideas and works, and Dr. K. Weissenberg, with whom I was able to discuss the final chapter.

I should also like to thank the Board of the B.R.P.R.A. for encouraging me to undertake this work, and for the provision of facilities for its execution.

L.R.G.T.

British Rubber Producers' Research Association, Welwyn Garden City

ACKNOWLEDGEMENTS

THE National Bureau of Standards (Washington) with N. Bekkedahl for Figs 1.3 and 1.6 and with L. A. Wood for Fig. 1.7; The American Chemical Society for Figs 2.2, 2.3, 2.4, 2.7, 2.8 and 2.11; The American Chemical Society with P. J. Flory for Figs 7.5 and 8.1 and with M. Shen for Figs 13.5 and 13.6; The Soviet Academy of Sciences with A. P. Aleksandrov for Fig. 1.4; The Institution of the Rubber Industry with L. Mullins for Fig. 1.5 and with G. Gee for Fig. 7.8; Applied Scientific Research (Martinus Nijhoff) for Fig. 1.9; Helvetica Chimica Acta for Fig. 2.1; The Chemical Society for Figs 1.10, 5.4, 5.5, 5.6, 5.7, 5.8, 5.9, 7.1, 7.10, 7.11, 7.12, 9.2, 9.3, 9.6, 9.6, 9.7 and 9.8; The Chemical Society with L. Mullins for Figs 5.11, 5.12, and 5.13, with G. Gee for Figs 7.2, 7.3, and 7.6, with D. W. Saunders for Figs 9.11. 9.12, 9.13, 9.14, and 9.15, and with G. Allen for Figs 13.1, 13.2, 13.3, and 13.10; The American Institute of Physics with L. A. Wood for Fig. 2.9, with E. Guth for Fig. 2.10 with R. F. Landel for Fig. 11.2, and with M. Shen for Fig. 13.4; John Wiley and Sons Ltd., with J. Scanlan for Fig. 4.5, with L. Mullins for Figs 6.13, 6.14, 7.7, 8.3, 8.5, and 8.6; with M. C. Morris for Fig. 6.14, with W. F. Watson for Fig. 8.4, with A. N. Gent for Fig. 9.16, with D. W. Saunders for Fig. 9.18, with S. Kawabata for Figs 11.3, 11.4, 11.8, and 11.9, and with R. G. Christensen for Fig. 13.9, I.P.C. Business Press Ltd. for Figs 7.13, and 13.8, and with G. Allen for Fig. 13.7; The Royal Society with D. W. Saunders for Figs 10.1, 10.8, 12.9, and 12.10, and with R. W. Ogden for Fig. 11.1; The Institute of Physics for Figs 9.9, 9.10, 10.5, 10.6, 10.7, 11.5, 11.6, 11.7, and 11.11; Marcel Dekker Inc. with S. Kawabata for Fig. 11.10.

CONTENTS

1.	GENERAL PHYSICAL PROPERTIES OF RUBB	ER
	1.1. What is a rubber?	1
	1.2. Chemical constitution of rubbers	3
	1.3. Early theories of rubber elasticity	6
	1.4. The kinetic theory of elasticity	7
	1.5. Cross-linking and vulcanization: network theory	11
	1.6. The glass–rubber transition	13
	1.7. Crystallization in raw rubber	16
	1.8. Crystallization in the stretched state	20
2.	INTERNAL ENERGY AND ENTROPY Tentrapi	
	CHANGES ON DEFORMATION	
	2.1. Stress-temperature relations * * * * * * * * * * * * * * * * * * *	24
	2.2. Thermodynamic analysis	28
	2.3. Application to experimental data	32
	2.4. Interpretation of thermoelastic data	34
	2.5. Thermal effects of extension	37
	2.6. Conclusion (Kan Klu: 3an) Wik	40
3.	THE ELASTICITY OF LONG-CHAIN	
	MOLECULES	
	3.1. Statistical properties of long-chain molecules	42
	3.2. Statistical form of long-chain molecule	43
	3.3. The randomly jointed chain	46
	3.4. Properties of Gaussian functions	48
	3.5. The distribution of <i>r</i> -values	50
	3.6. Equivalent random chain	53
	3.7. The entropy of a single chain	55
	3.8. The tension on a chain	57
4.	THE ELASTICITY OF A MOLECULAR	
	NETWORK	
	4.1. The nature of the problem	59
	4.2. Detailed development of the theory	60
	4.3. Significance of theoretical conclusions	64
	4.4. The principal stresses	65
	4.5. Significance of single elastic constant	67
	4.6. The elastic properties of a swollen rubber	68
	4.7. Development of the theory by James and Guth	71

	4.8.	Network imperfections: 'loose end' corrections	74		
	4.9.	The absolute value of the modulus	77		
_		TRUMPINE OF THE			
5.		ERIMENTAL EXAMINATION OF THE			
		TISTICAL THEORY	00		
	5.1.	Introduction	80		
		Particular stress-strain relations	81		
		Experimental examination of stress-strain relations	85		
		Deviations from theory: Mooney equation	95		
	5.5.	General conclusions	99		
6.	NON	N-GAUSSIAN CHAIN STATISTICS AND			
٠.		WORK THEORY			
	6.1.	Introduction have	101		
	6.2.	Statistical treatment of randomly jointed chain	102		
		Entropy and tension	106		
		Alternative derivation of tension on chain	108		
	6.5.	The exact distribution function	109		
		Application to real molecular structures	111		
		Non-Gaussian network theory	113		
	6.8.		122		
	6.9.	Possible influence of crystallization	123		
		The equivalent random link	124		
7	CWE	LLING PHENOMENA			
1.		Introduction	128		
	7.1.		129		
		Experimental data	131		
	7.4.		134		
		Significance of thermodynamic quantities	136		
		Statistical treatment of swelling	139		
		Comparison with experiment The swelling of cross-linked polymers	140		
	7.7.	Relation between swelling and modulus	140		
		The cohesive-energy density	147		
	7.9.	The dependence of swelling on strain	150		
		Experiments on swelling of strained rubber	155		
		Swelling under torsional strain	158		
	1.12.	Swelling under torsional strain	130		
8.	CROSS-LINKING AND MODULUS				
	8.1.	Introduction	160		
	8.2.	Early work	161		
	8.3.	The experiments of Moore and Watson and of Mullins	164		
	8.4.	Effect of entanglements [in teng mant] n 445/2	168		
	8.5.	Discussion and conclusion	170		

	9.	PHC	TOELASTIC PROPERTIES OF RUBBERS		
		9.1.	Refractive index and polarizability	174	
		9.2.	Optical properties of long-chain molecules	175	
		9.3.	The Gaussian network	178	
		9.4.	The effect of swelling	182	
		9.5.	The non-Gaussian network	182	
		9.6.	Measurement of birefringence	186	
		9.7.	Investigations on natural rubber	189	
		9.8.	The effect of the degree of cross-linking	195	
		9.9.	Polyethylene	198	
		9.10.	Optical properties of the monomer unit	202	
			The equivalent random link	204	
			The effect of swelling on stress-optical coefficient	206	
		9.13.	Temperature dependence of optical anisotropy	209	
1	10.		GENERAL STRAIN:		
		PHE	NOMENOLOGICAL THEORY		
	-	10.1.	Introduction	211	
		10.2.	The theory of Mooney	212	
		10.3.	Rivlin's formulation	214	
		10.4.	Pure homogeneous strain	218	
		10.5.	The general strain: early experiments	220	
		10.6.	The experiments of Rivlin and Saunders	223	
		10.7.	Interpretation of Mooney plots	225	
		10.8.	Molecular significance of deviations from statistical		
			theory	227	
	11.	ALT	ERNATIVE FORMS OF STRAIN-ENERGY		
		FUN	CTION		
		11.1.	Survey of alternative proposals	230	
		11.2.	Ogden's formulation	233	
		11.3.	The Valanis-Landel hypothesis	236	
		11.4.	Experimental examination of Valanis-Landel		
			hypothesis	238	
		11.5.	Form of the function $w'(\lambda)$	242	
		11.6.	Re-examination in terms of strain invariants	246	
	12.	LAR	GE-DEFORMATION THEORY: SHEAR		
	ANDTORSION				
		12.1.	Introduction: components of stress	252	
		12.2.	Stress components in simple shear	253	
			Torsion of a cylinder	258	
		12.4.	Generalization of preceding results	260	
		12.5.	Experimental verification	262	

CONTENTS

	12.6.	Further problems in torsion	265
	12.7.	Simultaneous extension, inflation, and shear of	
		cylindrical annulus ?	267
	12.8.	Application of Ogden formulation	269
13.	THEI	RMODYNAMIC ANALYSIS OF GAUSSIAN	
		WORK	
		Introduction	270
		Force-extension relation for Gaussian network	271
	13.3.		272
		Internal energy and entropy changes	274
		Measurements at constant volume	276
		Values of f_e/f	281
		Alternative experimental methods	283
		Theoretical analysis of torsion	287
	13.9.	Experimental data for torsion	291
	13.10.	Volume changes due to stress	293
		Experimental examination	294
	13.12.	Volume changes in torsion	296
	13.13.	Calorimetric determination of internal-energy	
		contribution to stress	297
	13.14.	Temperature dependence of chain dimensions	299
		Conclusion	300
RE	FERE	NCES	302
AU	тноі	R INDEX	307
SUI	BJECT	INDEX	309

GENERAL PHYSICAL PROPERTIES OF RUBBERS

1.1. What is a rubber?

THE original material of commerce known as rubber (or more precisely 'india-rubber') is obtained in the form of latex from the tree Hevea Braziliensis. The more expressive term 'caoutchouc', derived from the Maya Indian words meaning 'weeping wood', in reference to the exudation of the latex from a wound in the bark (Le Bras 1965), has been retained by the French, and transliterated into other European languages. The word rubber is derived from the ability of this material to remove marks from paper, to which attention was drawn by the chemist Priestley in 1770 (Memmler 1934, p. 3). In current usage the term rubber is not restricted to the original natural rubber, but is applied indiscriminately to any material having mechanical properties substantially similar to those of natural rubber, regardless of its chemical constitution. The more modern term elastomer is sometimes employed in relation to synthetic materials having rubber-like properties, particularly when these are treated as a sub-class of a wider chemical group. However, in the present work the more popular usage will be followed. It will generally be obvious from the context whether the word rubber is used in the general or in the more restricted sense; in cases where confusion might arise it will be sufficient to refer to natural or Hevea a.河流 电表流. rubber.

The reasons for this choice are not entirely verbal. It is at least equally justifiable from the scientific standpoint to define a rubber in terms of its physical properties as in terms of its chemical constitution. Indeed, in the present work, we shall be concerned very much more with those fundamental structural aspects in which all rubbers may be considered to be essentially the same than with the more detailed specific features in which they differ from one another. The emphasis will be placed mainly on rubber-like elasticity as a phenomenon associated with the rubber-like state of matter.



The most obvious and also the most important physical characteristic of the rubber-like state is of course the high degree of deformability exhibited under the action of comparatively small stresses. A typical force-extension curve for natural rubber is shown in Fig. 1.1; the maximum extensibility normally falls within

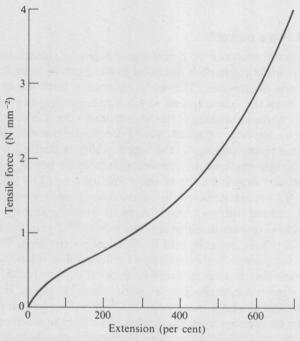


FIG. 1.1. Typical force-extension curve for vulcanized rubber.

the range 500–1000 per cent. The curve is markedly non-linear (i.e. Hooke's law does not apply), hence it is not possible to assign a definite value to Young's modulus except in the region of small strains. In this region its value (represented by the tangent to the curve at the origin) is of the order of $1\cdot0~\rm N~mm^{-2}$. These properties—high extensibility and low modulus—are to be contrasted with the properties of a typical hard solid (e.g. steel), for which the value of Young's modulus is $2\cdot0\times10^5~\rm N~mm^{-2}$ and the corresponding maximum elastic (i.e. reversible) extensibility about $1\cdot0$ per cent or less. There is thus an enormous difference between

rubbers on the one hand and ordinary hard solids (crystals, glasses, metals, etc.) on the other.

Thermoelastic effects

In addition to these familiar mechanical properties rubber also possesses a number of other less well-known properties, namely, the thermal or thermoelastic properties, which are of even greater scientific significance. The study of these properties dates from the beginning of the last century, when Gough (1805) made the following two observations, i.e.

(1) that rubber held in the stretched state, under a constant load,

contracts (reversibly) on heating; and

(2) that rubber gives out heat (reversibly) when stretched.

Gough's conclusions were confirmed some 50 years later by Joule (1859), who worked with the more perfectly reversible vulcanized rubber which had become available since the time when Gough's original experiments were carried out. The two effects referred to are usually known as the Gough-Joule effects. An example of the second, taken from Joule's publications, is reproduced in Fig. 2.10 (p. 38); this shows the rise of temperature due to the evolution of heat on stretching up to an extension of 100 per cent.

These thermoelastic effects are not peculiar to natural rubber, but are characteristic of the rubber-like state, being observed in a wide

variety of synthetic rubbery polymers.

1.2. Chemical constitution of rubbers

Natural rubber is essentially a hydrocarbon, whose constitution was established by Faraday (1826) to be consistent with the formula $(C_5H_8)_n$. The rubber exists in the latex in the form of small globules, having diameters in the range $0\cdot 1-1\cdot 0$ μ m, suspended in a watery liquid or serum, the concentration of the rubber being about 35 per cent. The rubber particles would coalesce, of course, were it not for a layer or sheath of non-rubber constituents, principally proteins, which is adsorbed on their surfaces and functions as a protective colloid. From this latex the solid rubber may be obtained either by drying off the water or by precipitation with acid. The latter treatment yields the purer rubber, since it leaves most of the non-rubber constituents in the serum.

Chemically, the rubber hydrocarbon is a polymer of isoprene (C₅H₈) built up in the form of a continuous chain having the

n. 空清. 经承

FIG. 1.2. The structure of the molecule of (a) Hevea rubber and (b) gutta-percha. A-B = isoprene unit. C = methyl group.

structure shown in Fig. 1.2. The succession of isoprene units in the chain is perfectly regular, with every fourth carbon atom in the chain carrying the methyl (CH₃) side-group. The presence of the double bond is very significant, since it is this that largely determines the chemical reactivity of the molecule and its ability to react with sulphur or other reagents in the vulcanization process. The double bond is also responsible for the susceptibility of the rubber molecule to oxidation or other degradative reactions leading to a deterioration of physical properties (aging). 7. 易志性、政志性

The structure of gutta-percha, the other natural polymer of isoprene, differs slightly but significantly from that of rubber. As will be seen from Fig. 1.2, the difference lies solely in the arrangement of the single C—C bonds with respect to the double bonds in the chain backbone. In rubber the single bonds lie on the same side of the double bond, forming the so-called cis-configuration, whilst in gutta-percha they lie on opposite sides of the double bond, giving the trans-configuration. One consequence of this difference is that gutta-percha crystallizes more readily than rubber; it is in fact crystalline at room temperature, becoming rubber-like only when heated above the crystal melting point, namely, 65 °C.

Although the two single bonds adjacent to the double bond remain permanently fixed in a single plane (whether in the cis- or trans-configuration), the remaining single bonds are not thus fixed but are subject to rotation out of the plane formed by neighbouring bonds, as will be discussed in detail later. The structural forms

† In these copolymers the respective monomer units occur in a random sequence along the chain.